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BaT₂As₂ single crystals (T = Fe, Co, Ni) and superconductivity upon Co-doping

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ABSTRACT

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1. Introduction

The discovery of superconductivity in a variety of compounds with square lattice sheets of Fe^{2+} in the structure has created great attention primarily due to their non-Cu²⁺ based origin. The structure types adopted by iron-based superconductors include those of ZrCuSiAs-type RFeAsO [1,2], and ThCr₂Si₂-type BaFe₂As₂ [3]. In the oxypnictide RFeAsO system, electron- and hole-doping has produced critical temperatures (T_c) as high as \approx 55 K in SmFe-AsO_{0.9}F_{0.1} [4], Gd_{1.8}Th_{0.2}FeAsO [5], and SmFeAsO_{0.85} [6]. In the oxygen-free compounds of AFe_2As_2 (A = Ca, Sr, Ba) system, holedoping has reached T_c values of 38 K by K-doping [7]. The superconductivity induced by electron-doping with cobalt in BaFe₂As₂ was reported by us earlier [8]. This demonstrates that in-plane disorder is highly tolerated in Fe-sheets, without creating localized moments. This scenario is consistent with our report on LaFeAsO system [2] and in contrast with the high T_c cuprates, as Zn^{2+} doping induces localized moments and destroys superconductivity. Due to the existing interest in the properties of ThCr₂Si₂-type BaT₂As₂ materials (T = transition metal), the thermodynamic and transport properties of single crystalline T = Fe, Co, and Ni samples are surveyed. Also, bulk superconductivity by Co-doping in BaFe₂As₂ single crystals is reviewed.

2. Results and discussion

The crystal structure and physical properties of BaFe₂As₂, BaCo₂As₂, and BaNi₂As₂ single crystals are sur-

veyed. BaFe₂As₂ gives a magnetic and structural transition at $T_N = 132(1)$ K, BaCo₂As₂ is a paramagnetic metal, while BaNi₂As₂ has a structural phase transition at $T_0 = 131$ K, followed by superconductivity

below $T_c = 0.69$ K. The bulk superconductivity in Co-doped BaFe₂As₂ below $T_c = 22$ K is demonstrated

by resistivity, magnetic susceptibility, and specific heat data. In contrast to the cuprates, the Fe-based sys-

tem appears to tolerate considerable disorder in the transition metal layers. First principles calculations

for BaFe_{1.84}Co_{0.16}As₂ indicate the inter-band scattering due to Co is weak.

In the preparation of crystals, high purity elements (>99.9%) were used and the source of the elements was Alfa Aesar. The single crystals of BaT₂As₂ were grown each out of its respective TAs binary where T = Fe, Co, or Ni [8-10]. Such self flux is preferred over other metal solvents such as Sn or Pb, as flux impurities can become incorporated in the crystals. For BaFe₂As₂ or BaCo₂As₂, a 1:5 ratio of Ba:FeAs or Ba:CoAs was heated for 15 h at 1180 °C under partial argon atmosphere. In both cases the ampoules were cooled at the rate of 2–4 °C/h, followed by decanting of flux by use of a centrifuge at 1090 °C. For Co-doped BaFe₂As₂, a ratio of Ba:FeAs:CoAs = 1:4.45:0.55 was heated to 1180 °C, and held for 10 h. The reaction was cooled at the rate of 3-4 °C/h, followed by decanting of FeAs flux at 1090 °C. For the growth of BaNi₂As₂ single crystals, a ratio of Ba:NiAs = 1:4 was heated in an alumina crucible for 10 h at 1180 °C under a partial atmosphere of argon. This reaction was cooled at the rate of 3 °C/h, followed by decanting of flux at 1025 °C. The typical crystal sizes from all batches were $\sim 6 \times 5 \times 0.2 \text{ mm}^3$ or smaller. The crystals were brittle, wellformed plates with the [001] direction perpendicular to the plane of the crystals. The BaCo₂As₂ and BaNi₂As₂ crystals were found to





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Fig. 1. Room temperature lattice constants for BaT_2As_2 with T = Fe, Co, and Ni, refined from X-ray powder diffraction data.

be highly air-sensitive. Electron probe microanalysis of a cleaved surface of the single crystal was performed on a JEOL JSM-840 scanning electron microscope using an accelerating voltage of 15 kV and a current of 20 nA with an EDAX brand energy-dispersive X-ray spectroscopy (EDS) device. EDS analyses on all parent crystals indicated Ba:T:As ratio of 1:2:2. EDS analyses on several crystals of the Co-doped BaFe₂As₂ indicated that 8.0(5)% of the Fe is replaced by Co in BaFe₂As₂. This composition is presented as BaFe_{1.84}Co_{0.16}As₂.

 BaT_2As_2 (T = Fe, Co, Ni) and Co-doped $BaFe_2As_2$ crystallize with the ThCr₂Si₂ structure-type [11,12] at room temperature, in tetrag-

onal space group symmetry I4/mmm (No. 139; Z = 2). The crystal structure is made up of $Ba_{0.5}^{2+}(TAs)^{-1}$ for the parents, with partial and random substitution of Co on Fe sites for BaFe_{1.84}Co_{0.16}As₂. At room temperature, the transition metal atoms lie on a perfect square net in the *ab*-plane of the tetragonal structure. The phase purity of the crystals was determined using a Scintag XDS 2000 2θ - 2θ diffractometer (Cu K α radiation). The cell parameters were refined using least squares fitting of the peak positions in the range 2θ from 10 to 90° using the Jade 6.1 MDI package. These are shown in Fig. 1. The relative decrease in *c*-lattice parameter is \sim 12% on going from BaFe₂As₂ to BaNi₂As₂. The increase in *a*-lattice parameter is smaller (Fig. 1). The refined lattice constants of BaFe_{1.84-} $Co_{0.16}As_2$ are a = 3.9639(4) Å and c = 12.980(1) Å. Cobalt doping results in a small decrease (0.3%) in the length of the BaFe₂As₂ caxis, while the value of *a*-axis is unchanged within experimental uncertainty. We have reported similar behavior in Co-doped LaFe-AsO [2].

Upon cooling, BaFe₂As₂ and BaNi₂As₂ undergo symmetry-lowering crystallographic phase transitions. For BaFe₂As₂, a transition at 132 K is associated with a tetragonal to orthorhombic (*Fmmm*) symmetry. Below the phase transition, the four equal Fe–Fe bonds at 280.2 pm are split into two pairs with 280.8 pm and 287.7 pm lengths [3]. For BaNi₂As₂, a tetragonal to orthorhombic first-order phase transition was suggested [13] below T_0 = 131 K only by analogy with AFe₂As₂ (A = Ba, Sr, Ca) compounds. However, we have recently found that the symmetry of the low temperature structure is triclinic ($P\bar{1}$) [10]. The reduction in symmetry below T_0 results in a distorted Ni network with short Ni–Ni contacts (2.8 Å) and longer Ni–Ni distances (3.1 Å). For BaCo₂As₂, the low temperature



Fig. 2. (a) For BaFe₂As₂ and along the two crystallographic directions, the temperature dependence of magnetic susceptibility in zero-field-cooled (*zfc*) forms measured at 1 T. The inset shows the average susceptibility, $(2\chi_{ab} + \chi_c)/3$, in the form of $d(\chi T)/dT$ peaking at T_N . (b) For BaFe_{1.84}Co_{0.16}As₂ in 20 Oe along *ab*-plane, the temperature dependence of susceptibility in *zfc* and field-cooled (*fc*) forms below 30 K, assuming χ value of the perfect diamagnetism. The inset is the $\chi(T)$ in *zfc* forms along in 1 T. (c) For BaNi₂As₂ and along the two crystallographic directions, the *zfc* temperature dependence of magnetic susceptibility in 1 T. Inset of (c) is the temperature dependence of χ_{ab}/χ_c for BaCo₂As₂.

diffraction was not studied as no structural transition is expected based on the physical properties, summarized below.

DC magnetization was measured as a function of temperature using a Quantum Design Magnetic Property Measurement System. Fig. 2a shows the measured magnetic susceptibility (χ) in zerofield-cooled (zfc) form for BaFe₂As₂ in 1 T along c- and ab-crystallographic directions. At room temperature, $\chi_c \approx \chi_{ab} \approx 7 \times 10^{-4}$ cm³ mol⁻¹. The susceptibility decreases linearly with decreasing temperature, and drops abruptly below ~135 K, with $\chi_c > \chi_{ab}$ below. The polycrystalline average of the susceptibility data are presented as the Fisher's $d(\chi T)/dT$ [14] versus temperature (Fig. 2a, inset) to infer $T_{\rm N}$ = 132 K. Shown in Fig. 2b is the temperature dependence of $4\pi\chi_{eff}$ measured along *ab*-plane at an applied field of 20 Oe for BaFe_{1.84}Co_{0.16}As₂, where χ_{eff} is the effective magnetic susceptibility after demagnetization effect correction, as described in ref. [8]. The zfc $4\pi\chi_{eff}$ data decreases rapidly for temperatures below ~22 K and quickly saturates to 1. This indicates that the system is indeed fully shielded and the substitution of Co on Fe-site is more homogenous than the use of other dopants such as fluorine in LaFeAsO_{1-x} F_x [2] or potassium in Ba_{1-x} K_x Fe₂As₂ [7] for which the transitions are broadened. In addition, cobalt is much easier to handle than alkali metals or fluorine. The fc $4\pi \gamma_{\rm eff}$ saturates at much smaller value (<0.01), reflecting strong pinning due to the Co dopant. As seen in the inset of Fig. 2b, γ for BaFe_{1.84-} $Co_{0.16}As_2$ varies smoothly between \sim 22 K and 300 K, indicative of the disappearance of magnetic ordering. For BaNi₂As₂, the magnetic susceptibility is anisotropic (Fig. 2c) χ is roughly temperature independent and presents only a small drop at $T_0 \approx 31$ K and a rise below ~ 12 K. For BaCo₂As₂, the magnetic susceptibility decreases with increasing temperature and is anisotropic with $\chi_{ab}/\chi_c \approx 0.7$ (Fig. 2c, inset). At 1.8 K, $\chi_c = 5.4 \times 10^{-3} \text{ mol}^{-1} \text{ Oe}^{-1}$ and $\chi_{ab} = 3.8 \times 10^{-3} \text{ mol}^{-1}$ 10^{-3} mol⁻¹ Oe⁻¹. The field dependent magnetization for T = Ni and Co are linear at 1.8 K (data not shown).

Temperature dependent electrical resistivity was performed on a Quantum Design Physical Property Measurement System (PPMS). The resistivity was measured in the *ab*-plane (ρ_{ab}), as described in Ref. [8]. While both BaFe₂As₂ and BaFe_{1.84}Co_{0.16}As₂ show metallic behavior, the resistivity for BaFe_{1.84}Co_{0.16}As₂ is smaller than for the parent compound (Fig. 3a). At room temperature, $\rho_{ab} = 0.50 \text{ m}\Omega \text{ cm}$ for BaFe₂As₂ and $\rho_{ab} = 0.32 \text{ m}\Omega \text{ cm}$ for BaFe_{1.84}-Co_{0.16}As₂. For BaFe₂As₂ below ~135 K, there is a sharp step-like drop. This feature is best manifested in the derivative of resistivity, $d\rho/dT$, giving a peak at 132(1) K. For BaFe₂As₂ the residual resistivity ratio RRR (= $\rho_{300 \text{ K}}/\rho_{2 \text{ K}}$) is 4. The magnitude of the resistivity at 2 K and 8 T is higher than the zero-field value for BaFe₂As₂ (top inset of Fig. 3a). This corresponds to a positive magnetoresistance of

 $(
ho_{\rm 8T}ho_{\rm 0})/
ho_{\rm 0}$ = 27.7% at 2 K. For BaFe_{1.84}Co_{0.16}As_2, an abrupt drop in ρ_{ab} is observed below 22 K. The onset transition temperature for 90% fixed percentage of the normal state value is $T_c^{\text{onset}} = 22$ K. The transition width in zero-field is $\Delta T_c = T_c$ $(90\%) - T_c$ (10%) = 0.6 K. The ΔT_c value is much smaller than that reported for LaFe_{0.92}Co_{0.08}AsO (2.3 K) [15], and for LaFeAsO_{0.89}F_{0.11} (4.5 K) [2]. The resistive transition shifts to lower temperatures by applying 8 T, and transition width becomes wider (1.3 K), characteristic of type-II superconductivity (Fig. 3a, bottom inset). Ba-Co₂As₂ and BaNi₂As₂ show metallic behavior (Fig. 3b). At room temperature these materials are more conductive $(\rho_{ab} \approx 0.07 \text{ m}\Omega \text{ cm})$ than BaFe₂As₂. The residual resistivity ratio RRR is \sim 8 for each, illustrating good crystal quality. For BaCo₂As₂, ρ_{ab} decreases with decreasing temperature, varying approximately linearly with T^2 below ~60 K (bottom inset of Fig. 3b). By fitting this data using $\rho_{ab}(T) = \rho_{ab}(0 \text{ K}) + AT^2$, we obtain the residual resistivity of 5.7 $\mu\Omega$ cm, and $A = 2.2 \times 10^{-3} \mu\Omega$ cm/K². Comparison with the Sommerfeld coefficient below, gives a Kadowaki-Woods relation $A/\gamma^2 = 0.5 \ \mu\Omega \times 10^{-5} \ \mu\Omega \ \text{cm mol Co}^2 \ \text{K}^2/\text{mJ}^2$ consistent with attributing the temperature dependence of the resistivity to electron-electron scattering. For BaNi₂As₂, there is a sharp increase at T_0 in ρ followed by a continuous decrease with decreasing temperature. The superconductivity downturn in ρ_{ab} is at ~1.5 K (top inset of Fig. 3b).

Specific heat data, $C_{p}(T)$, were also obtained using the PPMS, via the relaxation method below 200 K. For BaFe₂As₂ (Fig. 4a), specific heat gives a broad anomaly, peaking at 132(1) K. For BaFe_{1.84-} Co_{0.16}As₂ (inset of Fig. 4a), there is a specific heat jump below $T_{\rm c}$ = 22 K, confirming bulk superconductivity. There are no specific heat features for BaCo₂As₂ (Fig. 4b) suggesting no phase transition up to 200 K, while for BaNi₂As₂ there is a sharp, pronounced transition at $T_0 = 132$ K (Fig. 4b). For BaNi₂As₂, the temperature dependence of C/T in several fields for H/|ab is shown in Fig. 4b, inset. In zero-field, there is a sharp anomaly at $T_c = 0.69$ K with a jump ΔC = 12.6 mJ/(K mol). Taking the value of the Sommerfeld coefficient at T_c , this gives a ratio of $\Delta C/\gamma_n T_c = 1.38$, roughly comparable to a value of 1.43 predicted by weak-coupling BCS theory and confirms bulk superconductivity. As the magnetic field increases, the transition becomes broader and shifts to lower temperatures. The upper critical field is $H_{c2}(0) = 0.069$ T and the coherence length is $\xi_{GL}(0) \approx 691$ Å, as described in ref. [10]. Below ~7 K, C/T has a linear T^2 dependence for BaT₂As₂ (Fig. 4c). The fitted Sommerfeld coefficient, γ , for BaFe₂As₂ is 6.1 mJ/(K² mol) [or 3.0 mJ/(K² mol) Fe)]. BaCo₂As₂ gives $\gamma = 41.6 \text{ mJ}/(\text{K}^2 \text{ mol})$ [or 20.8 mJ/(K² mol Co)], consistent with a Fermi liquid plus phonon contribution. The Wilson ratio $R_w = \pi^2 k_B^2 \chi/(3\mu_B^2\gamma)$ for BaCo₂As₂ is ~7 from χ_{ab} and ~10



Fig. 3. Temperature dependence of resistivity for BaT_2As_2 measured in *ab*-plane. (a) $\rho_{ab}(T)$ for $BaFe_2As_2$ and $BaFe_{1.84}Co_{0.16}As_2$. Top inset is the behavior measured for $BaFe_2As_2$ at 0 T and 8 T (blue line). (b) $\rho_{ab}(T)$ for $BaCo_2As_2$ and $BaNi_2As_2$. Top inset is $\rho(T)$ below 2 K for BaNi_2As_2. Bottom inset depicts $\rho(T^2)$ for $BaCo_2As_2$ between 1.8 K and 70 K, and linear fit below ~60 K. (For interpretation of the references in colour in this figure legend, the reader is referred to the web version of this article.)



Fig. 4. Temperature dependence of the specific heat between 1.8 K and 200 K for (a) $BaFe_2As_2$, and (b) $BaCo_2As_2$ and $BaNi_2As_2$. The inset of (a) is the C/T vs. T^2 for $BaFe_{1.84}Co_{0.16}As_2$, depicting anomaly below $\sim T_c$. The inset of (b) is C/T vs. T for $BaNi_2As_2$ below 0.8 K at 0 and 50 G. The T_c was found using the onset criterion demonstrated by the dashed line for the zero-field data. (c) C/T vs. T^2 for BaT_2As_2 (T = Fe, Co, Ni) and a linear fit between 1.8 K and ~ 7 K.

from χ_c (measured at 2 K). These values well exceed unity for a free electron system and indicate that the system is close to ferromagnetism [9]. For BaNi₂As₂ the normal state electronic specific heat coefficient, γ_{n} , is 13.2 mJ/(K² mol).

First principles calculations for the layered Fe–As superconductors show that the electronic states near the Fermi energy, i.e. those electronic states that are involved in superconductivity, are predominantly derived from Fe d states, and that there is only modest hybridization between the Fe d and As p states [16]. The electronic structures of the corresponding Co and Ni compounds



Fig. 5. Projected electronic density of states on a per atom basis for a supercell of composition BaFe_{1.75}Co_{0.25}As₂ showing that the Fe and Co have similar density of states indicating moderate scattering and near rigid band behavior following Ref. [8].

are very similar in a rigid band sense. Specifically, if one compares the electronic structures of BaFe₂As₂ [17] BaCo₂As₂ [8,9] and Ba-Ni₂As₂ [18,19] one finds that the band structures and density of states are closely related between the different compounds, the main difference being that the Fermi level is shifted according to the electron count. This is important because it means that the alloys will remain metallic with only moderate alloy scattering. Supercell calculations, motivated by our discovery of superconductivity in Ba(Fe,Co)₂As₂ [8] showed that this is in fact the case. Fig. 5 shows projections of the electronic density of states onto the Fe sites and the Co site in a supercell of composition $BaFe_{1.75}Co_{0.25}As_2$, following Ref. [8]. As may be seen, the shapes of the projections onto Fe and Co are remarkably similar, consistent with modest scattering and rigid band alloy behavior. Calculations for BaMn₂As₂ show different behavior, however [20]. In this compound, there is strong spin dependent hybridization between Mn and As. This implies that Mn substitution in BaFe₂As₂ or the other compounds would likely lead to strong scattering and carrier localization.

3. Conclusions

The shrinking of the *c*-lattice parameter is observed from T = Fe to Ni in BaT₂As₂, at room temperature. For BaFe₂As₂, the tetragonal to orthorhombic *Fmmm* distortion also involves a second-order-like spin-density-wave magnetic transition below $T_{\rm N}$. In contrast to this broader transition, for BaNi₂As₂ there is a sharp, first-order, pronounced feature at T_0 = 131 K, associated with a reduction of the lattice symmetry from tetragonal to triclinic $P\overline{1}$. The superconducting critical temperature for BaNi₂As₂ is at T_c = 0.69 K. Ba-Co₂As₂, midway between BaFe₂As₂ (d⁶) and BaNi₂As₂ (d⁸), shows no evidence of structural or magnetic transitions. BaCo₂As₂ shows

evidence of strong fluctuations and is predicted to be in close proximity to a quantum critical point.

The bulk nature of the superconductivity for 8.0(5)% Co-doped BaFe₂As₂ single crystal is confirmed by narrow transition widths in $\rho(T)$, $\chi(T)$, and also the anomaly in C(T). Based on band structure calculations, the superconductivity of BaFe₂As₂ appears to be different from BaNi₂As₂. The Ni-phase can be understood within the context of conventional electron-phonon theory, while the high T_c Fe superconductor is not described in this way. Supercell calculations show that BaFe₂As₂ will remain metallic with only moderate Co-alloy scattering. Thus the BaT_2As_2 compounds, T = Fe, Co, and Ni, while all metallic, show a remarkably wide range of interesting properties, including both conventional and unconventional superconductivity, spin-density-wave antiferromagnetism, and proximity to ferromagnetism. Furthermore, results so far indicate that these materials can be continuously tuned between these states using alloving of these three T elements. This provides an avenue for exploring in detail the interplay between the different states.

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